

Formation of *p-n* junction by plasma spraying technique to enhance the photocatalytic activity of TiO₂

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The mineralization of toxic organic pollutants by photoreaction, using semiconductors such as TiO₂, CdS, SnO₂, WO₃, SiO₂, ZrO₂, ZnO, Nb₂O₅, Fe₂O₃, SrTiO₃ etc., has attracted extensive attention after the discovery of the Fujishima and Honda phenomenon [1–3]. Among all the oxide semiconductors that have been reported, anatase titanium dioxide is an excellent photocatalyst because of its high stability against photocorrosion and favorable band-gap energy (3.2 eV) [2]. To improve the photocatalytic efficiency, it is of great importance to inhibit the recombination of the photo-generated electrons and holes. To separate the photo-generated charges spatially, one approach is to apply composite films or powders such as TiO₂/WO₃, TiO₂/Nb₂O₅ [4] etc.

Generally, the photocatalytic activity increases with increasing specific surface. Therefore, micro-powders are often applied as photocatalysts because the specific surface area is larger than that of a membrane. But in practical applications, micro-powder is very difficult to reclaim after photocatalytic reaction. In order to avoid this kind of technical problem, a number of methods have been used to form TiO₂ films, including wet chemical processing (e.g., sol-gel, screen printing), vapor processing techniques (e.g., CVD, PVD) and thermal oxidation of Ti [3].

The coating formation speed and bonding strength are very low and it is difficult to produce large surface coatings by chemical processes such as sol-gel and CVD. However, the plasma spraying technique is an economical and versatile fabrication process to produce large surface coatings with almost unlimited types of materials. The coating thickness, texture and bonding strength can be controlled easily through spraying parameters, nature of the powders and substrate state [5]. Moreover, a plasma sprayed deposit is formed by a stream of molten droplets impacting on the substrate followed by flattening, rapid solidifying and cooling process. The individual molten (or half-molten) droplets spread to form thin lamellae, the stacking of which constitutes the deposit [6]. Therefore, it is easy to deposit composite coatings by spraying technique if the feedstock powders are composite materials.

The composite substances may react with each other and then produce new compounds for the heating process; and the interface between two kinds of compounds is very large because the droplets spread to form very thin lamellae (micron order). In the present research, the plasma spraying technique was used to deposit composite photocatalytic coatings using composite TiO₂-Fe₃O₄ powder. To investigate the composition of sprayed coatings in detail, thin lamella (splat) was also formed.

TiO₂ and TiO₂-Fe₃O₄ coatings and corresponding splats were formed using a Plasmadyne SG-100 torch with the conditions shown in Table I.

The microstructure, distribution of elements and compositions of the sprayed samples were analyzed by electron probe surface roughness analyzer (ERA-8800FE) (Elionix Co. Ltd., Japan) and X-ray diffraction using Cu K_α radiation ($\lambda = 1.5406 \text{ \AA}$) with a graphite crystal monochromator (JDX3530, Jeol, Japan).

The photocatalytic activity of the sprayed coatings was evaluated through the degradation of acetaldehyde under 1 mW/cm² ultraviolet light irradiation. The photoelectrochemical characteristics of plasma sprayed coatings in 0.1 N NaOH solution were studied through a three-electrode cell system under 30 mW/cm² xenon light irradiation. The details are described elsewhere [3, 7].

The photocatalytic degradation efficiency (defined by the inclination of the exponential degradation curve) of the TiO₂-Fe₃O₄ coating is approximately two times higher than that of the TiO₂ coating as shown in Fig. 1. According to the X-ray diffraction patterns, the TiO₂ coating consists of anatase and rutile TiO₂ phase. In the TiO₂-Fe₃O₄ coatings, the *n*-type TiO₂ semiconductor particle reacted with the Fe₃O₄ particle and

TABLE I Plasma spraying parameters

Ar gas pressure (MPa)/flow (slpm)	0.42/58
He gas pressure (MPa)/flow (slpm)	0.21/9
Arc current (A)	600
Arc voltage (V)	30
Spraying distance (mm)	70

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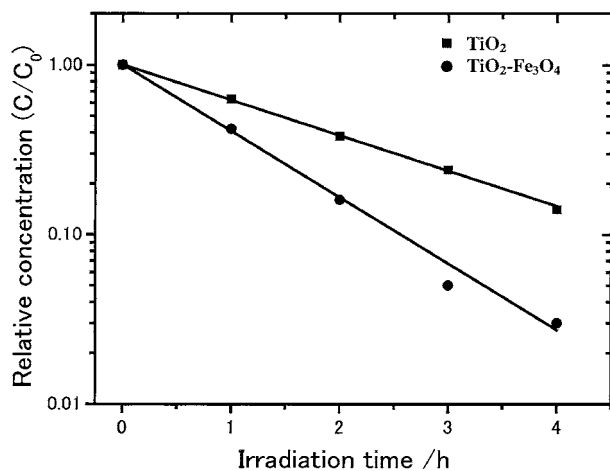


Figure 1 The photocatalytic decomposition characteristics of the acetaldehyde by plasma sprayed TiO_2 and $\text{TiO}_2\text{-Fe}_3\text{O}_4$ coatings.

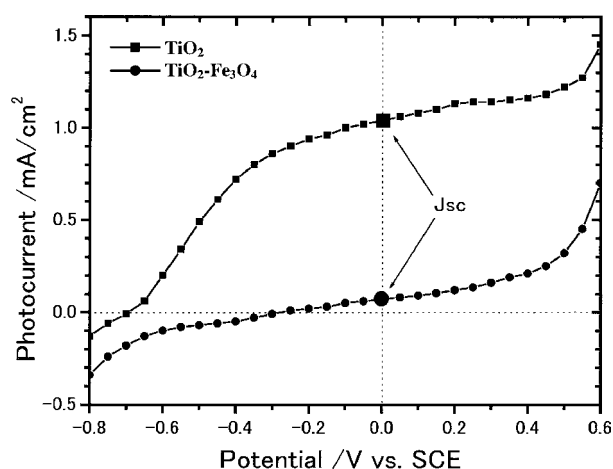


Figure 2 The photocurrent-potential curves of plasma sprayed TiO_2 and $\text{TiO}_2\text{-Fe}_3\text{O}_4$ electrodes.

concurrently produced p -type FeTiO_3 , and consequently produced a p - n junction in the interface between the flattened TiO_2 lamella and FeTiO_3 compound. For the formation of a micro p - n junction, which may spatially separate the photo-generated electrons and holes as in a Si solar cell, the photocatalytic activity was improved. The existence of a p - n junction was clearly proved by the photocurrent-potential measurement and EDAX analysis results. The photocurrent density (photocurrent value (J_{sc}), with potential equal to 0 V) of sprayed $\text{TiO}_2\text{-Fe}_3\text{O}_4$ coating was much lower than that of TiO_2 coating (Fig. 2), but the bubble (from water photolysis) formation speed from the $\text{TiO}_2\text{-Fe}_3\text{O}_4$ electrode is about three times quicker than that from the TiO_2 . That is to say, the $\text{FeTiO}_3/\text{TiO}_2$ islands act as micro-photoelectrolysis cells in a short circuit configuration in the $\text{TiO}_2\text{-Fe}_3\text{O}_4$ coating, and the initiated electron cannot flow to the cathode (Pt) through the output circuit and this results in a low photocurrent. The 3-D view, plane view and line A analysis results of $\text{TiO}_2\text{-Fe}_3\text{O}_4$ splat are shown in Fig. 3. The thickness of the splat is not greater than $1 \mu\text{m}$. The circumference is thicker than the central region, which may result from the high impacting speed of the particle on the substrate. The Ti and Fe elements dis-

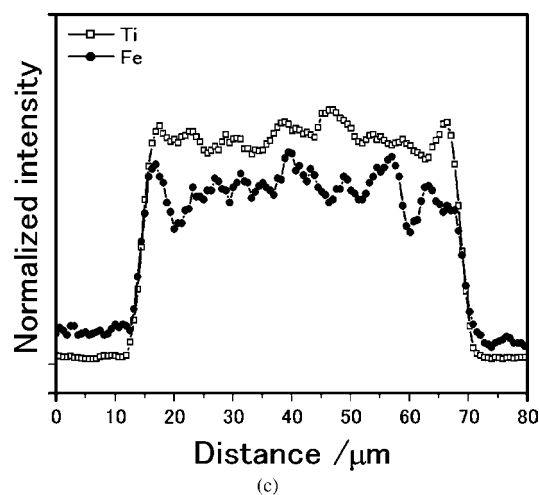
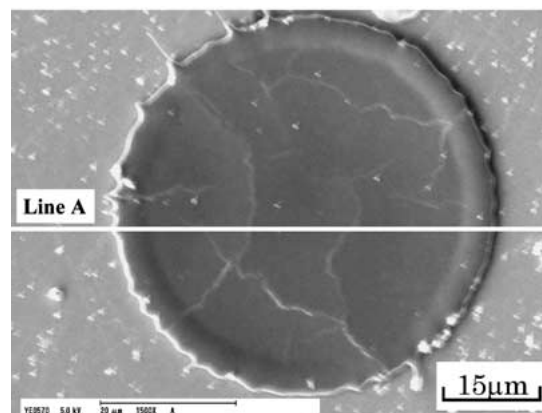
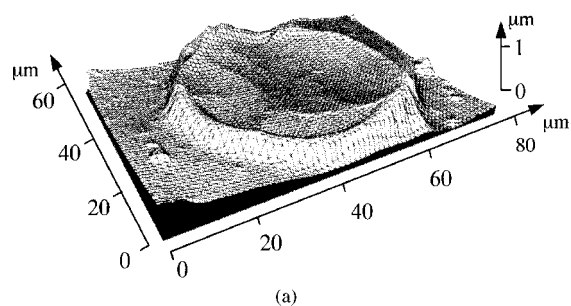


Figure 3 The 3-D view (a), plane view (b) of $\text{TiO}_2\text{-Fe}_3\text{O}_4$ splat, and Ti, Fe elements distribution (c) of line A illustrated in plane view.

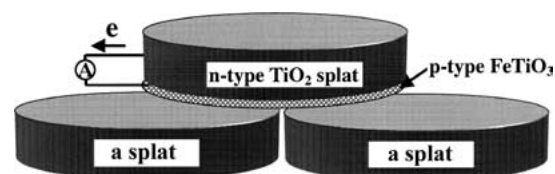


Figure 4 A proposed p - n junction formation model in the $\text{TiO}_2\text{-Fe}_3\text{O}_4$ coating.

tribute uniformly as illustrated in Fig. 3c, and it can be reasonably inferred that numerous p - n junctions were formed in it. The $\text{Ti}/(\text{Fe} + \text{Ti})$ ratio in the $\text{TiO}_2\text{-Fe}_3\text{O}_4$ splat approximates to 0.88, which is in good agreement with the ratio for the feedstock powder and sprayed coating. According to the above results, a p - n junction formation model in plasma sprayed $\text{TiO}_2\text{-Fe}_3\text{O}_4$ coating is illustrated in Fig. 4. The micro- TiO_2 and FeTiO_3

constitutes innumerable micro-cells. From these investigations, a promising approach for manufacturing high performance photocatalytic coatings by plasma spraying technique, using its special advantages, has been suggested.

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*Received 29 May
and accepted 9 July 2003*